A Robust Modification of SiO₂ Nanoparticles by Poly(2-hydroxyethylmethacrylate) via Surface-Initiated Atom Transfer Radical Polymerization

Duy Trinh Nguyen¹, Thuong Nhan Phu Nguyen¹, Duy Chinh Nguyen¹, Van Thi Thanh Ho², Md. Rafiqul Islam³, Kwon Taek Lim³ and Long Giang Bach^{1,*}

¹Nguyen Tat Thanh University, 300A Nguyen Tat Thanh, District 4, Ho Chi Minh City, Vietnam

Received: 24 July 2018;

16

17

18

19

20

21

22

23

24

25

26

27

28 29

30

31

32

33

34

Accepted: 25 September 2018;

Published online: 31 December 2018;

AJC-19205

35

37

38

39

40

41

42

43

44

45

46

47

48

49

50

51

52

53

54

55

An effortless and efficient method for the alteration of SiO_2 nanoparticles by poly(2-hydroxyethylmethacrylate) (PHEMA) has been developed. Initially, a strategic atom transfer radical polymerization (ATRP) initiator was anchored to SiO_2 nanoparticles surface *via* coupling reaction followed by surface-initiated atom transfer radical polymerization of HEMA successfully afforded chemically grafted PHEMA to SiO_2 nanoparticles (PHEMA-g-SiO₂). We used XPS, FT-IR and EDS analyses for confirming the covalent immobilization of PHEMA onto SiO_2 nanoparticles. TGA/DSC and zeta potentials measurements were used for determining the thermal and surface charge features of the nanocomposites, respectively.

Keywords: Surface functionalization, SiO₂ Nanoparticles, Polymer chains, Poly(2-hydroxyethylmethacrylate).

INTRODUCTION

There is a growing interest from modern fields such as electronics, optical, biomedicine and renewable energy generation for development multifunctional nanomaterials. Nanostructured multiphase materials could be conformed to obtain multi-functionality, because various characteristics of different materials can be combined to develop a wide range of fascinating features [1-8]. Efficiency and performance of products are enhanced using engineering of the interfaces in multiphase materials. The combination of functional polymers and surface alteration of nanoparticles is a vibrant area of current attention because of their fantastic electronics, optical, magnetic and catalytic features. Surface functionalization generally requires appropriate synthetic methods that offer exact control over material features. Several methods have been developed to modify nanoparticles by polymers which significantly improved their mechanical properties [9-21].

Nowadays, surface-initiated atom transfer radical polymerization (SI-ATRP) technique is recognized as a potential and flexible methods for effective grafting of accurately identi-

fied polymers, star polymers, block copolymers, graft copolymer and hyperbranched polymers from a variety of solid surfaces [22-32]. The key advantage of this technique is the complete control over the molecular weight (M_n, M_w), polydispersity (PDI) and chain-end functionality of grafted polymer. It also provides high graft density of polymer allowing multifunctional applications. Moreover, the ATRP is a widely practiced method now that it does not ask to required conditions and is accepting of functional species and contaminants that are harmful to another "living" polymerization techniques.

Recently, synthesis of polymer grafted hybrid nanoparticles having reactive pendant functionalities, such as hydroxyl, carboxylic acid or epoxide groups, has gained substantial interest as building blocks for the fabrication of advanced nanostructured devices. Biocompatible poly(2-hydroxyethyl methacrylate) is an intriguing material for biology and medicine fields as in medication carrier, dental, contact lenses, antithrombotic equipment and soft-tissue replacement. The essential features of poly(2-hydroxyethylmethacrylate) (PHEMA) such as mechanical features, thermal steady, colloidal steady, moisture content and affinity to specific biological molecules can be

This is an open access journal, and articles are distributed under the terms of the Creative Commons Attribution-NonCommercial 4.0 International (CC BY-NC 4.0) License, which allows others to copy and redistribute the material in any medium or format, remix, transform, and build upon the material, as long as appropriate credit is given and the new creations are licensed under the identical terms.

²Hochiminh University of Natural Resources and Environment, 236B Le Van Sy, Ward 1, Tan Binh District, Ho Chi Minh City, Vietnam ³Pukyong National University, Busan 608-737, Republic of Korea

^{*}Corresponding author: Fax :+ 84 28 39404759, Tel : + 84 96 9294297; E-mail: blgiangntt@gmail.com

338 Nguyen et al. Asian J. Chem.

adjusted to response with new requests. The features of PHEMA can be adjusted by the combination of nanomaterials and commanded by controlling the ingredient of inorganic or organic components, nanotexture, essence of inorganic and/or organic segments, etc. [33-37].

56

57

58

59

60

61

62

63

64

65

66

67

68

69

70

71

72

73

74

75

76

77

78

79

80

81

82

83

84

85

86

87

88

89

90

91

92

93

94

Among inorganic nanoparticles, silica spheres (SiO₂ nanoparticles) have excellent mechanical strength and thermal stability. They are also non-toxic, easy to functionalize and can be created controllably by hydrolysis of organosilicate at room temperature, which makes them one of the most commonly used inorganic cores. Consequently, polymers graft silica nanocomposites have received considerable attention in recent years because of high demand for optical equipment, strengthened elastomers and plastics, bioactive glass, chemical sensors, biology and medicine sensors and other types [38-40].

In this work, the exploration of an effortless and efficient procedure for precise surface alteration of SiO₂ nanoparticles with poly(2-hydroxyethylmethacrylate) (PHEMA) will not only reveal the surface properties of nanoparticles but also open up new prospects for versatile applications. We demonstrate a facile strategy for the preparation of clear identified SiO₂/PHEMA nanocomposites via SI-ATRP. The macroinitiator (SiO₂-Br) was first prepared by one step reaction using 3-glycidyloxypropyltrimethoxysilane (GOTMS) and 2-bromo-2-methylpropionic acid (BMPA). To obtain PHEMA-g-SiO₂ core-shell type nanocomposites, the surface of SiO₂ nanoparticles is modified by surface-initiated atom transfer radical polymerization of 2-hydroxyethylmethacrylate (HEMA).

EXPERIMENTAL

The SiO₂ nanoparticles with the mean size of 108 nm were prepared using Stöber technique [41] with surface area of ca. 115 m²/g as measured by BET method. Purified 2-hydroxyethyl methacrylate (HEMA) was produced before hand by flowing a neutral alumina column to evict the inhibitor. NH₄OH (25 wt. % aqueous solution), tetraoctylammonium bromide (TOAB), tetraethylorthosilicate (TEOS), 3-glycidyloxypropyltrimethoxysilane (GOTMS), 2-bromo-2-methylpropanoic acid (BMPA), triethylamine (TEA), N,N,N',N',N"-pentamethyldiethylenetriamine (PMDETA), CuBr, tetrahydrofuran (THF), N,N-dimethylformamide (DMF) and all solvents were used as introduced.

Synthesis of strategic initiator (GOTMS-BMPA): 3-Glycidyloxypropyltrimethoxysilane (GOTMS) (2.00 g), 2-bromo-2methylpropanoic acid (BMPA) (1.42 g) and THF (6 mL) were placed in a pyrex tube. The pipe was emptied and back-filled with nitrogen three times, then secured and held in a thermostatically with the oil bath at 70 °C and stirred very quickly under N₂ atmosphere for 1 day. Lastly, all volatile materials was removed under vacuum and marked as GOTMS-BMPA. 102 The ¹H NMR spectrum of GOTMS-BMPA (CDCl₃, 400 MHz, ppm): $\delta = 4.15$ (CH₂-CH(OH)-CH₂-O-CO-, 2H, dd); 3.67-3.71 (CH₂-CH(OH)-CH₂-O-CO-, 1H, m); 3.62 (CH₂-CH(OH)-CH₂-O-CO-, 2H, d); 3.53 ((CH₃O)₃Si-, 9H, s); 3.45-3.47 (CH₂O-(CH₂)₃Si-, 2H, t); 1.91 (-(CH₃)₂-Br, 6H, s); 1.90 (CH₂-CH(OH)-CH₂-O-CO-, 1H, d); 1.67 (SiCH₂CH₂CH₂O, 2H, quint); 0.65 (SiCH₂CH₂CH₂O, 2H, d).

95

96

97 98

99

100

101

103

104

107

109

110

111

113

118

119

120

122

124

125

128

133

134

Synthesis of PHEMA-g-SiO₂ nanocomposites via SI-ATRP technique

Synthesis of SiO₂-Br macroinitiators: Dissolved 2.0 g of SiO₂ nanoparticles in 20 mL toluene and then added 1.0 g GOTMS-BMPA and 5 mL triethylamine (TEA) to create a 114 uniform suspension solution. Then, the mixture was strongly 115 stirred at 50 °C under N₂ flow for 1 day and subsequently, cooled down to 35 °C and cleaned in the order with toluene and dichloromethane. Next, the macroinitiator SiO₂-Br was dried at 40 °C under vacuum for 1 day.

Synthesis of PHEMA-g-SiO₂ nanocomposites by SI-**ATRP method:** Mixed 2-hydroxyethylmethacrylate (HEMA, 2 g), SiO₂-Br (0.2 g), PMDETA (0.024 mL), CuBr (0.0166 g) in 2 mL of THF using a Teflon-covered propulsion rod in a 25 mL circle bottle decorated with a restore condenser. The mixture was removed the gas with N₂ by three freeze-pumpthaw cycles and held in an oil tank with the temperature of 80 °C and stirred constantly. Then, the bottle was cooled to 35 °C. Next, diethyl ether was used to precipitate the mixture and the product was separated with centrifugation. The crude product 129 was cleaned and had impurities removed using THF multiple 130 times. Lastly, a vacuum oven was used to dry PHEMA-g-SiO₂ nanocomposites for a duration of 1 night at 40 °C. The synthetic route for SI-ATRP technique to prepare PHEMA-g-SiO₂ using SiO₂-Br macroinitiators is depicted in **Scheme-I**.

$$\begin{array}{c} -O \\ -O \\ -O \\ -O \\ \end{array} \\ \begin{array}{c} O \\ -O \\ \end{array} \\$$

Scheme-I: Synthetic protocol for the preparation of PHEMA-g-SiO₂ nanocomposites via SI-ATRP technique

150

151

152

154

155

157

158

159

161

162

165

166

167

168

170

179

180

182

186

187

190

191

192

193

194

195

197

198

199

200

201

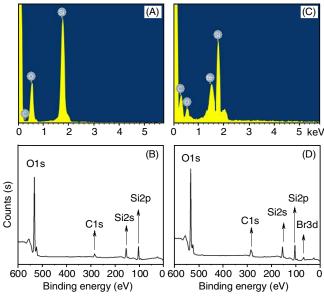
202

203

135 Characterizations: BOMEM Hartman & Braun spectrometer was used to determine FT-IR. Elemental analysis and 136 137 morphological analysis of the hybrids was performed using 138 FE-SEM photos attached with an EDX spectrometer (Hitachi JEOL- JSM-6700F). For TGA, Perkin-Elmer Pyris 1 analyzer (USA) was utilized. The DSC measurements were conducted 140 using a Perkin Elmer calorimeter (DSC6200). Surface compo-142 sition of nanocomposites was investigated using XPS (X-ray 143 photoelectron spectroscopy) (Thermo VG Multilab 2000). GPC 144 (gel permeation chromatography) analysis was performed using 145 an Agilent 1200 Series equipped with PLgel 5 µm MIXED-C columns, with N,N-dimethylformamide at 30 °C and calibration was performed by PMMA standards. The zeta potential for the 148 surface charge properties of samples was carried out using a Zeta Plus 90 analyzer. 149

RESULTS AND DISCUSSION

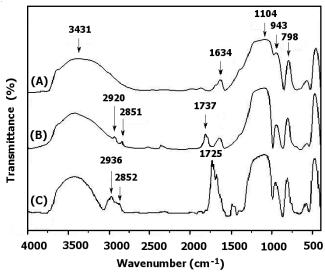
The strategic initiators (GOTMS-BMPA) were attached on the surface of SiO₂ nanoparticles in one step through ligandexchanging response among the -OH species on the surface of SiO₂ nanoparticles and trimethoxyliane groups of GOTMS-BMPA to produce SiO₂-Br macroinitiators. Fig. 1 showed the surface chemical composition of functionalized SiO₂ nanoparticles through EDS and XPS analyses. The EDX scan revealed the attendance of Si, O and C chemical components in SiO₂ nanoparticles as displayed in Fig. 1A. Fig. 1C showed that signals of Si, O and C elements affected the vast-scan XPS spectrum of SiO₂ nanoparticles surface. GOTMS-BMPA was attached on the surface of SiO₂ nanoparticles that causes a condensation response to form a stable initiator monolayer. Fig. 1B showed that clear signal of bromine atoms on surfaces of SiO₂-Br macroinitiators via EDS analysis. To assert the presence of ATRP initiators on SiO₂ nano-surfaces, SiO₂-Br macroinitiators was also analyzed XPS as shown in Fig. 1D. Expectedly, in the scan, we found that the primary peak component, nominated to O1s, and the small peak component, nominated to C1s, at the binding energy are of 533.6 and 285.0 eV, respectively. In addition to



(A, B) EDS and (C, D) XPS wide-scan spectra of (A, C) the SiO₂ NPs, (B, D) SiO2-Br macroinitiators

this, peaks at the binding energy of 154.8 eV (Si2s) and 103.7 eV (Si2p) were also found. In particular, the characteristic Br3d 171 peak as sighted at 69.2 eV, which verifies the presence of Br signals on SiO₂ nanoparticles surfaces. The EDS and XPS results 173 confirm the success of the formation of SiO₂-Br macroinitiators. 174

The FT-IR was used to study the surface chemical bonding of modified SiO₂ nanoparticles. Characteristically, asymmetric stretching of Si-O-Si is associated to the absorption band, detected at 1104 cm⁻¹. The broad absorption band centered around 3431 cm⁻¹ is assigned to -OH species on the surface of SiO₂ nanoparticles (Fig. 2A). Four absorption bands at 2920 cm⁻¹ (C-H stretching of -CH₃), 2851 cm⁻¹ (C-H stretching of =CH₂), 1737 cm⁻¹ (C=O stretching) and 1114 cm⁻¹ (Si-O-Si stretching which overlapped C-O-C stretching) (Fig. 2B) indicated the interaction between GOTMS-BMPA and the surface-bound OH groups of SiO₂ nanoparticles. The surface initiated ATRP of HEMA from GOTMS-BMPA enclosed SiO₂ nanoparticles surface was performed in the attendance of CuBr/PMDETA catalyst system to give PHEMA-g-SiO₂ nanocomposites. The successful synthesis of core-shell textured PHEMA-g-SiO₂ nanocomposites was verified by respective spectral and microanalyses. Fig. 2C displayed FT-IR spectrum of PHEMA-g-SiO₂, the broad absorption band at 3647-3022 cm⁻¹ is assigned for the O-H stretching vibration and growth in intensity at 2936, 2851 cm⁻¹ is due to the C-H stretching vibration. The characteristic absorption band appeared at 1725 cm⁻¹ shows C=O stretching vibration band. The typical bands at 1225-1180 cm⁻¹ assigned for the stretching of -C-O- (in ester species). The wide absorption bands at 800-500 cm⁻¹ assigned for Si-O-Si stretching which is generally frequent in SiO₂ associated forms, and the band sighted at 1634 cm⁻¹ indicates the attendance of physically adsorbed aqua in the SiO₂ forms. FT-IR analyses suggested that PHEMA was covalently attached with SiO₂ nanoparticles via SI-ATRP.



FT-IR spectra of (A) SiO₂ NPs, (B) SiO₂-Br macroinitiators, (C) PHEMA-g-SiO₂ nanocomposites

Fig. 3 showed the results of thermogravimetric analysis 204 from which the number of organic components anchored to SiO₂ nanoparticles is determined. From TGA analysis, the initial and final depravity temperature of the samples were determined.

205 206 340 Nguyen et al. Asian J. Chem.

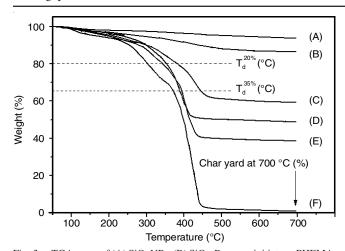


Fig. 3. TGA scans of (A) SiO₂ NPs, (B) SiO₂-Br macroinitiators, PHEMAg-SiO₂ after polymerization for (C) 8 h, (D) 16 h, and (E) 24 h and (F) cleaved PHEMA from PHEMA-g-SiO₂ nanocomposites

209

210

211

212

213

215

216

217

218

219

220

221

222

223

224

225

226

227

228

Visually, by increasing the temperature from 50 to 700 °C, the sample of SiO₂ nanoparticles loses approximately 5.1 % of their total weight. This is due to the removal of aqua molecules adsorbed on the outer and the liberation of textured aqua caused by the bonded -OH species. Thermogravimetric analysis (Fig. 3) shows that the amount of grafted GOTMS-BMPA was calculated as to be ca. 10.7 %. In temperature region from 285 to 430 °C, PHEMA-g-SiO₂ nanocomposites were decomposed, which is basically because of the breakdown of grafted PHEMA on SiO₂ nanoparticles. It is sighted that the weight decline of nanocomposites changed from 40.2 to 61.2 % based on the polymerization time from 8 to 24 h. The results suggested a reasonable level of functionalization of SiO₂ nanoparticles by PHEMA.

The condition in which SI-ATRP polymerization occurred and molecular weight of grafted PHEMA were determined by separating the grafted polymers from PHEMA-g-SiO₂ nanocomposites using HF aqueous solution. The GPC analysis was performed to determine M_n and PDI of separated PHEMA. In Fig. 4, M_n and PDI of grafted PHEMA was plotted in the two vertical axes and the overall monomer conversions were displayed in the horizontal axis. The M_n of PHEMA was found to be proportional with increasing monomer conversion. The cleaved PHEMA

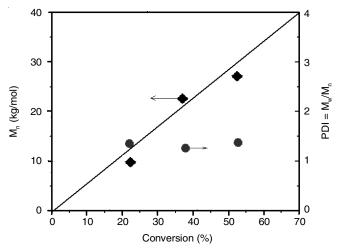
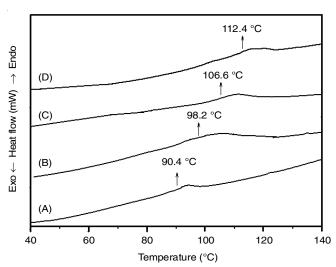


Fig. 4. M_n and PDI of the cleaved PHEMA from PHEMA-g-SiO₂ nanocomposites at different polymerization time (8, 16 and 24 h)

at polymerization time of 8, 16 and 24 h resulted in M_n of 10.5, 22.7 and 27.8 kg/mol with relatively narrow PDI of 1.37, 1.26 and 1.41, respectively, suggesting that SI-ATRP of HEMA from SiO₂ nanoparticles surface took place in a controlled way. 234

Thermal property of separated PHEMA and PHEMA-g-SiO₂ nanocomposites was further examined by DSC. The glass transition temperature (T_g) of cleaved PHEMA is observed at 237 90.4 °C. On the other hand, the T_g of nanocomposites at polymerization time of 8, 16 and 24 h were found to be 98.2, 106.6 and 112.4 °C, respectively. It was observed that T_g of PHEMAg-SiO₂ enhanced by the high grafting of PHEMA. The T_g value 241 of grafted PHEMA is higher by ca. 8-20 °C than that of the equivalent neat PHEMA form (Fig. 5). The increase of T_g can be explained in such a way that the confinement of one head of PHEMA series on SiO₂ nanoparticles would limit both the motion and vibration of whole series that eventually happened in strong interaction as well as inherent high modulus of SiO₂ nanoparticles resulted in increased T_g.



DSC curves of (A) cleaved PHEMA, PHEMA-g-SiO₂ nanocomposites at polymerization time of (B) 8 h, (C) 16 h and (D) 24 h, respectively

The electrical property of particles surfaces regarding the 249 zeta potential (ζ) is an essential factor for applications in biomedical, industrial and environmental areas. The ζ of modified 251 SiO₂ nanoparticles in suspension was determined using a Brookhaven Zeta Plus 90 analyzer. All the dispersions were adjusted to the desired pH with HCl (0.25 M) and NaOH (0.25 M). ζ vs pH bends for SiO₂ nanoparticles and PHEMA-g-SiO₂ nanocomposites are plotted (Fig. 6). For SiO₂ nanoparticles, negative zeta voltage was found all the whole examined pH region. The surface charge density of SiO₂ nanoparticles suspensions was observed to be decreased as the pH of suspension was increased and the isoelectric point (pI) was reached around at pH 3.0. In opposition, ζ vs. pH bends for the PHEMA- g-SiO₂ nanocomposites indicated the survival of pI at ca. pH = 6 and positive zeta voltage of situated above to + 40 mV was attained at low pH. So, this results given proof that the covalent modification of SiO₂ nanoparticles by PHEMA significantly increased the surface charge behavior of unmodified SiO₂ nanoparticles which can be used as a non-viral vector for transferring genetic materials (gene, DNA, RNA) across cellular membranes.

257

231

236

240

243

244

245

246

247

248

267

268

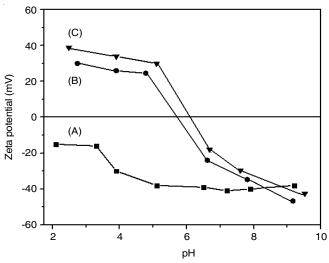


Fig. 6. Zeta potential vs. pH curves for (A) SiO₂ NPs and PHEMA-g-SiO₂ at polymerization time of (B) 8 h and (C) 24 h, respectively

269 Conclusion

270 Surface modification of SiO₂ nanoparticles by poly(2-271 hydroxyethylmethacrylate (PHEMA) was accomplished to 272 afford PHEMA-g-SiO₂ nanocomposites via a facile SI-ATRP method. Following the anchoring of a sacrificial initiator GOTMS-274 BMPA to the surface of SiO_2 nanoparticles, chemically bonded 275 PHEMA-g-SiO₂ nanocomposites were synthesized by SI-ATRP 276 of 2-hydroxyethylmethacrylate (HEMA) from the surface of 277 SiO₂ nanoparticles employing grafting from strategy as confir-278 med by FT-IR analyses. The thermal steady of grafted PHEMA 279 was dramatically improved compared to pure PHEMA as 280 suggested by TGA and DSC studies. The controlled nature of 281 polymerization was evidenced by the linear increase of M_n of 282 the separated PHEMA from PHEMA-g-TiO₂ nanocompo-sites 283 with monomer conversion and the relatively low PDI (< 1.5). 284 The zeta potential of functionalized SiO₂ nanoparticles 285 indicated that surface modification by PHEMA using SI-ATRP 286 is a fantastic strategy for manipulating surface electrostatic 287 character of SiO₂ nanoparticles. This facile chemical strategy 288 holds a great promise for surface engineering of nanoparticles 289 in material science and nanotechnology to devise a cascade of 290 promising materials for diverse applications.

ACKNOWLEDGEMENTS

This work was supported by Nguyen Tat Thanh University, Vietnam.

CONFLICT OF INTEREST

The authors declare that there is no conflict of interests regarding the publication of this article.

REFERENCES

- M. Dabrowski, P. Lach, M. Cieplak and W. Kutner, *Biosens. Bioelectron.*, 102, 17 (2018); https://doi.org/10.1016/j.bios.2017.10.045.
- L. Peponi, D. Puglia, L. Torre, L. Valentini and J.M. Kenny, *Mater. Sci. Eng. Rep.*, 85, 1 (2014); https://doi.org/10.1016/j.mser.2014.08.002.

- M. Wang, X. Wang, M. Chen, Z. Yang and C. Dong, *Chin. J. Catal.*, 37, 1037 (2016);
 - https://doi.org/10.1016/S1872-2067(16)62477-4
- W. Salim and W.S.W. Ho, Curr. Opin. Chem. Eng., 8, 76 (2015); https://doi.org/10.1016/j.coche.2015.03.003.
- J. Song and J. Jang, Adv. Colloid Interface Sci., 203, 37 (2014); https://doi.org/10.1016/j.cis.2013.11.007.
- 6. M. Mir, N. Ahmed and A. Rehman, *Colloids Surf. B Biointerfaces*, **159**, 217 (2017);
 - https://doi.org/10.1016/j.colsurfb.2017.07.038.
- S. Kassim and M. Pemble, *Asian J. Chem.*, 30, 1617 (2018); https://doi.org/10.14233/ajchem.2018.21280.
- M. Santhiago, P.S. Garcia and M. Strauss, Curr. Opin. Green Sustain. Chem., 12, 22 (2018); https://doi.org/10.1016/j.cogsc.2018.04.009.
- L.G. Bach, X.T. Cao, M.R. Islam, H.G. Kim and K.T. Lim, *J. Nanosci. Nanotechnol.*, 15, 5897 (2015); https://doi.org/10.1166/jnn.2015.10438.
- L.G. Bach, X.T. Cao, V.T.T. Ho, M.R. Islam and K.T. Lim, *Mol. Cryst. Liq. Cryst.*, 618, 120 (2015); https://doi.org/10.1080/15421406.2015.1076315.
- X.T. Cao, L.G. Bach, M.R. Islam and K.T. Lim, *Mol. Cryst. Liq. Cryst.*, 618, 111 (2015); https://doi.org/10.1080/15421406.2015.1076305.
- L.G. Bach, B.T.P. Quynh, M.R. Islam and K.T. Lim, *J. Nanosci. Nanotechnol.*, 16, 12856 (2016); https://doi.org/10.1166/jnn.2016.13651.
- L.G. Bach, X.T. Cao, B.T.P. Quynh, V.T.T. Ho and K.T. Lim, Mol. Cryst. Liq. Cryst., 644, 183 (2017); https://doi.org/10.1080/15421406.2016.1277478.
- L.G. Bach, B.T.P. Quynh, N.T. Thuong and V.T.T. Ho, *Mol. Cryst. Liq. Cryst.*, **644**, 175 (2017); https://doi.org/10.1080/15421406.2016.1277476.
- L.G. Bach, B.T.P. Quynh and V.T.T. Ho, J. Nanosci. Nanotechnol., 17, 4127 (2017); https://doi.org/10.1166/jnn.2017.13381.
- C. Leopold, T. Augustin, T. Schwebler, J. Lehmann, W.V. Liebig and B. Fiedler, J. Colloid Interface Sci., 506, 620 (2017); https://doi.org/10.1016/j.jcis.2017.07.085.
- 17. G. Yang, X. Li, Y. He, J. Ma, G. Ni and S. Zhou, *Prog. Polym. Sci.*, **81**, 80 (2018);
- https://doi.org/10.1016/j.progpolymsci.2017.12.003.
- S. Mallakpour and E. Khadem, *Prog. Polym. Sci.*, 51, 74 (2015); https://doi.org/10.1016/j.progpolymsci.2015.07.004.
- S. Agbolaghi, S. Abbaspoor and F. Abbasi, *Prog. Polym. Sci.*, **81**, 22 (2018);
- https://doi.org/10.1016/j.progpolymsci.2017.11.006.

 20. A. Sosnik, J. das Neves and B. Sarmento, *Prog. Polym. Sci.*, **39**, 2030
- (2014); https://doi.org/10.1016/j.progpolymsci.2014.07.010.
- S. Kumar, M. Sarita, M. Nehra, N. Dilbaghi, K. Tankeshwar and K.-H. Kim, *Prog. Polym. Sci.*, 80, 1 (2018); https://doi.org/10.1016/j.progpolymsci.2018.03.001.
- Y. Zhao, L. Wang, A. Xiao and H. Yu, Prog. Polym. Sci., 35, 1195 (2010); https://doi.org/10.1016/j.progpolymsci.2010.05.002.
- P. Krys and K. Matyjaszewski, Eur. Polym. J., 89, 482 (2017); https://doi.org/10.1016/j.eurpolymj.2017.02.034.
- J. Ran, L. Wu, Z. Zhang and T. Xu, Prog. Polym. Sci., 39, 124 (2014); https://doi.org/10.1016/j.progpolymsci.2013.09.001.
- F. Seidi, H. Salimi, A.A. Shamsabadi and M. Shabanian, *Prog. Polym. Sci.*, 76, 1 (2018);
 - https://doi.org/10.1016/j.progpolymsci.2017.07.006.
- P. Król and P. Chmielarz, *Prog. Org. Coat.*, 77, 913 (2014); https://doi.org/10.1016/j.porgcoat.2014.01.027.
- Y. Liu and C.E. Hobbs, *Polymer*, 135, 25 (2018); https://doi.org/10.1016/j.polymer.2017.12.001.
- P. Polanowski, K. Halagan, J. Pietrasik, J.K. Jeszka and K. Matyjaszewski, *Polymer*, 130, 267 (2017); https://doi.org/10.1016/j.polymer.2017.10.011.
- J.M. Kubiak, J. Yan, J. Pietrasik and K. Matyjaszewski, *Polymer*, 117, 48 (2017); https://doi.org/10.1016/j.polymer.2017.04.012.

342 Nguyen et al. Asian J. Chem.

- 30. M. Chen, L. Qin, Y. Liu and F. Zhang, *Microporous Mesoporous Mater.*, **263**, 158 (2018);
 - https://doi.org/10.1016/j.micromeso.2017.12.019.
- L. Huang, M. Liu, L. Mao, D. Xu, Q. Wan, G. Zeng, Y. Shi, Y. Wen, X. Zhang and Y. Wei, *Appl. Surf. Sci.*, 412, 571 (2017); https://doi.org/10.1016/j.apsusc.2017.04.026.
- S. Kumar, P. Karfa, R. Madhuri and P.K. Sharma, *J. Phys. Chem. Solids*, 116, 222 (2018); https://doi.org/10.1016/j.jpcs.2018.01.038.
- M. Kubo, T. Kondo, H. Matsui, N. Shibasaki-Kitakawa and T. Yonemoto, *Ultrason. Sonochem.*, 40, 736 (2018); https://doi.org/10.1016/j.ultsonch.2017.08.011.
- Z. Sadakbayeva, M. Dušková-Smrcková, A. Šturcová, J. Pfleger and K. Dušek, Eur. Polym. J., 101, 304 (2018); https://doi.org/10.1016/j.eurpolymj.2018.02.035.
- V.V. Filipovic, B.D. Bozic Nedeljkovic, M. Vukomanovic and S.L. Tomic, *Polym. Test.*, 68, 270 (2018); https://doi.org/10.1016/j.polymertesting.2018.04.024.

- S. Sundararajan, A.B. Samui and P.S. Kulkarni, *React. Funct. Polym.*, 130, 43 (2018); https://doi.org/10.1016/j.reactfunctpolym.2018.05.012.
- S. Wu, W. Du, Y. Duan, D. Zhang, Y. Liu, B. Wu, X. Zou, H. Ouyang and C. Gao, *Acta Biomater.*, 75, 75 (2018); https://doi.org/10.1016/j.actbio.2018.05.046.
- H. Chen, X. Zhang, P. Zhang and Z. Zhang, Appl. Surf. Sci., 261, 628 (2012); https://doi.org/10.1016/j.apsusc.2012.08.071.
- S. Mallakpour and M. Naghdi, *Prog. Mater. Sci.*, 97, 409 (2018); https://doi.org/10.1016/j.pmatsci.2018.04.002.
- Sh. Ammar, K. Ramesh, I.A.W. Ma, Z. Farah, B. Vengadaesvaran, S. Ramesh and A.K. Arof, *Surf. Coat. Technol.*, 324, 536 (2017); https://doi.org/10.1016/j.surfcoat.2017.06.014.
- L.G. Bach, M.R. Islam, Y.T. Jeong, H.S. Hwang and K.T. Lim, *Mol. Cryst. Liq. Cryst.*, 565, 78 (2012); https://doi.org/10.1080/15421406.2012.692262.